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**APPLICATION
FOR
UNITED STATES
LETTERS PATENT**

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**FOR: WIRING STRUCTURE FOR
SEMICONDUCTOR DEVICE**

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WIRING STRUCTURE FOR SEMICONDUCTOR DEVICE

The present application is based on Japanese patent application No.2003-010044, the entire contents of which are
5 incorporated herein by reference.

BACKGROUND OF THE INVENTION

FIELD OF THE INVENTION

This invention relates to a wiring structure of
10 semiconductor device and particularly to a wiring structure suitable for VLSI and USLI.

DESCRIPTION OF THE RELATED ART

The speeding-up of silicon (Si) semiconductor is
15 performed by reducing the device size to increase the integration density of transistors formed on the single-crystal Si substrate according to Moore's Law. At present, the production of 0.13 μm rule device has started. In 2005, it is planned to use ultra fine wiring with a thickness of less
20 than 0.10 μm . A reduction in wiring width and wiring interval due to the ultra fine process causes an increase in electrical resistance of wiring and in inter-wiring capacity and, therefore, the wiring signal propagation speed decreases according as the ultra fine process makes progress. It is
25 recognized especially after the 0.13 μm rule that the problem of wiring delay adversely affects the operation speed of device. In the future development of Si semiconductor, the conversion of wiring material to new one is indispensable for increasing the wiring signal propagation speed since it is difficult to

increase it by using the conventional ultra fine process.

The wiring delay can be reduced by using a wiring material with a resistivity lower than Al that is conventionally used as wiring material. In 1997, IBM corp., USA proposed using
 5 copper (Cu) wiring. The resistivity of bulk Cu is $1.7 \mu\Omega \cdot \text{cm}$ while the resistivity of bulk Al is $2.7 \mu\Omega \cdot \text{cm}$.

However, in making Cu wiring, since the vapor pressure of Cu etching product is low and therefore the semiconductor has to be heated to a high temperature, the process of Al wiring
 10 cannot be used that, as shown in FIG.1A, Al thin film 2 is formed on the entire semiconductor substrate 1 and then a wiring region is covered with resist 3 and then the unnecessary Al uncovered is removed by etching. Therefore, another process of making Cu wiring is proposed that, as shown in FIG.1B, a trench or
 15 viahole 11 with wiring width is formed in semiconductor 10 before forming Cu film 5 and then Cu atom is embedded therein and then unnecessary Cu part 6 is removed by polishing such as CMP (chemical mechanical polishing) (this process is called Damascene process). Further, since, in Cu wiring, it is
 20 necessary to prevent the diffusion of Cu atom to the Si semiconductor, a barrier layer 4 needs to be formed between the Si semiconductor and Cu (See e.g., Miki Moriyama et al., "Future Fabrication Techniques for Cu Wires used in Si ULSI Devices", Materia Japan, Vol.39, No.11, pp.901-908 (2000), and Kazuhide
 25 Abe et al., "High Reliable Cu Damascene Interconnects with Cu/Ti/TiN/Ti Layered Structure", Oki-Denki-Kenkyu-Kaihatsu, No.184, Vol.67, No.3, pp.65-68, (October, 2000)). Also, since Cu needs to be completely embedded into the trench or viahole, the thickness of Cu film must be reduced according as the wiring

width decreases.

However, in forming the barrier layer by Damascene process, it is difficult to uniformly form the barrier layer in the trench or viahole. Furthermore, the barrier layer needs
5 to have a thickness greater than 10 nm at thinnest portion in order to function as barrier layer for preventing the diffusion. As a result, the barrier layer must be thick to some extent and the resistivity increases due to the thickness. Still furthermore, the existence of barrier layer interferes with the
10 reduction of wiring interval. Thus, a technique for forming Cu wiring without using the barrier layer has been researched.

On the other hand, from the viewpoint of the resistivity of Cu as wiring material, when the wiring width is reduced to 0.10 μm , that width becomes equal to mean free path of electron
15 and, therefore, there is concern that the resistivity of Cu wiring material increases. This is because an increase in resistivity caused by electron scattering becomes significant according as the film thickness and average grain diameter decrease. Therefore, it is necessary to increase the average
20 grain diameter in order to lower the resistivity of Cu wiring.

Furthermore, in case of Cu wiring, different from the case of Al wiring, there is concern that some crystal grains are preferentially grown when being kept at room temperature after the film formation to produce a nonuniform grain distribution
25 and the reliability lowers due to dispersion of resistivity caused by the nonuniform grain distribution. Although the growth of large crystal grains contributes to a reduction in resistivity from the viewpoint of electrical resistance, it is necessary to control the microscopic structure of Cu wiring in

consideration of both increasing and equalization of grain diameter in order to produce a nano-scale Cu wiring with low resistivity and high reliability.

5 The inventors have found that the growth of crystal grain depends on intrinsic strain (energy assist given to Cu film by the substrate) to be generated at the interface of Si substrate and Cu film. The invention is invented in researching the lowering of resistivity caused by increasing of crystal grain diameter, based on the growth process of Cu crystal grain in
10 consideration of materials interposed between Si substrate and Cu film.

SUMMARY OF THE INVENTION

It is an object of the invention to provide a wiring
15 structure for semiconductor device to offer low resistivity and high reliability.

According to the invention, a wiring structure for semiconductor device, comprises:

a wiring layer that includes copper as main component;
20 and

a crystal grain promotion layer that promotes enlargement in a crystal grain of the wiring layer.

The crystal grain composing the wiring layer is enlarged by the existence of crystal grain promotion layer, and therefore
25 a nano-scale Cu wiring with low resistivity can be realized.

In the invention, it is preferred that the wiring structure for semiconductor device has been composed as follows (1) to (5).

(1) The crystal grain of the wiring layer satisfies a

relation of $D > 10 \times L$ where D is an average grain diameter of crystal grain to be enlarged finally in the wiring layer and L is mean free path of electron. By satisfying that relation, the enlarged crystal grain of wiring layer comes close to single
 5 crystal and the resistivity of wiring layer is thereby close to that of bulk.

(2) The crystal grain promotion layer is disposed between a semiconductor or dielectric film and the wiring layer.

(3) The crystal grain layer has a good contact with the
 10 semiconductor or dielectric film and the wiring layer and is of a material that has a low reactivity to the semiconductor or dielectric film and the wiring layer. Thereby, a wiring structure for semiconductor device to offer low resistivity and high reliability can be obtained.

15 (4) The crystal grain layer is of high melting point metal, or nitride or carbide of the high melting point metal.

(5) The crystal grain layer is of a material selected from the group of titanium, tantalum, titanium nitrides, titanium carbides, tantalum nitrides and tantalum carbides.

20 Such high melting point metal, or nitride or carbide of the high melting point metal has effect of promoting enlargement in crystal grain. The high melting point metal is metal with a melting point of 150°C or higher. It is desirably titanium (Ti), tantalum (Ta), zirconium (Zr), hafnium (Hf), vanadium (V),
 25 tungsten (W), niobium (Nb) or molybdenum (Mo). Especially titanium (Ti) and tantalum (Ta) have significant effect.

By being thus composed, a small number of enlarged Cu crystal grains are grown at the initial stage of room temperature grain growth, (111) face of Cu is preferentially

grown in the process of growth, and the enlarged Cu crystal grains are grown having an average grain diameter seven to eight times the wiring width W . With the average grain diameter satisfying $D > 10 \times L$, the resistivity of wiring layer can be lowered to that of bulk. Thus, a nano-scale Cu wiring with low resistivity and high reliability can be realized.

BRIEF DESCRIPTION OF THE DRAWINGS

The preferred embodiments according to the invention will be explained below referring to the drawings, wherein:

FIGS.1A and 1B are cross sectional views showing the conventional method of making a wiring structure;

FIG.2A is surface observation (SIM) images by FIB showing the behavior of room temperature grain growth in a wiring structure of the invention;

FIG.2B is surface observation (SIM) images by FIB showing the behavior of room temperature grain growth in a wiring structure of comparative example;

FIG.3A is cross section observation (SIM) images by FIB showing the behavior of room temperature grain growth in the wiring structure of the invention;

FIG.3B is cross section observation (SIM) images by FIB showing the behavior of room temperature grain growth in the wiring structure of comparative example;

FIG.4A is a surface observation image by FIB after heat treatment in the wiring structure of the invention;

FIG.4B is a surface observation image by FIB after heat treatment in the wiring structure of comparative example; and

FIG.5 shows resistivity changes before and after heat

treatment in the wiring structures of the invention and the comparative example.

DETAILED DESCRIPTION OF THE PREFERRED EMBODIMENTS

5 A wiring structure in the preferred embodiment of the invention will be described below. The evaluation items are resistivity measurement by four-point probe, surface/cross section observation (SIM image) by focused ion beam (FIB), and crystal structure analysis by X-ray diffraction (XRD). Si
10 substrate is used as the semiconductor, Si_3N_4 is formed on the Si substrate in order to observe the crystal grain of Cu film, and TiN is used as crystal grain promotion layer.

 The Si substrate is cleansed by acetone and IPA, and then it is laid in a chamber of sputtering apparatus. Then, the
15 chamber is highly vacuumed to 4×10^{-9} Torr, Ar of 8.5 CCM and N_2 of 0.5 CCM are introduced therein, and Si_3N_4 of 60 nm thick is formed thereon using Si as target at a pressure of 5×10^{-3} Torr. Then, the chamber is highly vacuumed to 4×10^{-9} Torr, Ar of 8.5 CCM and N_2 of 0.5 CCM are introduced therein, and TiN,
20 crystal promotion layer, of $0.1 \mu\text{m}$ thick is formed thereon using Ti as target at a pressure of 5×10^{-3} Torr. Then, the chamber is highly vacuumed to 4×10^{-9} Torr, Ar of 10.0 CCM is introduced therein, and Cu, wiring layer, of $1.0 \mu\text{m}$ thick is formed thereon using Cu as target at a pressure of 6×10^{-3} Torr.
25 A comparative example without TiN layer is prepared. The property evaluation is conducted such that the behavior of grain growth at room temperature and after heat treatment of 2.5 hours at room at 350°C in argon (Ar) atmosphere is evaluated with respect to the above evaluation items.

The result of XRD measurement, though the spectrum is not shown herein, proves that, in the invention with TiN layer, comparing to the comparative example, the intensity of (111) of Cu increases and (111) face is grown preferentially.

5 FIGS.2A and 2B are the surface observation (SIM) images by FIB to show the behavior of room temperature grain growth, and FIGS.3A and 3B are the cross section observation images. FIGS.2A and 3A are those of the invention and FIGS.3A and 3B are those of the comparative example. Viewing from the results,
10 it is proved that in the invention large grains are less at the initial stage of room temperature grain growth and the grains are preferentially grown in the process of growth. The large grain of the invention has an average diameter seven to eight times that of the comparative example at the saturation stage,
15 and this average diameter is 10 times or more the mean free path (39 nm) of electron. Therefore, it is confirmed that TiN has a crystal grain promotion function. Furthermore, the time required to reach the saturation stage of the invention is reduced to about one third of that of the comparative example.
20 Therefore, it is also confirmed that the crystal grain promotion function is significant.

 FIGS.4A and 4B are surface and cross section observation images by FIB after heat treatment. FIG.4A is that of the invention and FIG.4B is that of the comparative example. Both
25 in the invention and the comparative example, the average diameter of large grain does not appear to be significantly grown after heat treatment. However, according as the microscopic grain region decreases, the average grain diameter increases and the diameters are equalized. As a result, it is

found that the average grain diameter of the invention becomes about twelve times that of the comparative example. Also, it is found that the average grain diameter at room temperature grain growth highly influences the grain diameter distribution after heat treatment.

FIG.5 shows resistivity changes before and after heat treatment. As shown, the resistivity of the invention is lower than that of the comparative example both before heat treatment and after heat treatment.

Although in the above embodiment of the invention the Si substrate is used as semiconductor, the same results are obtained in case of the Si substrate with SiO_2 formed thereon. It is obvious that the same effects will be obtained in case of having Si semiconductor layer (including impurity-doped layer and multiple layers) formed on the Si substrate and in case of further having SiO_2 film on that semiconductor layer. With regard to D (=average grain diameter of crystal grain) $> 10 \times L$ (mean free path of electron), the average grain diameter of Cu may have no upper limit and it is desirable that the diameter becomes infinitely large to give a single crystal. Thereby, the scattering of electron due to crystal grain field can be reduced. It is desirable that the heat treatment is conducted in the range of 100 to 400 $^{\circ}\text{C}$. Because, in this range, the invention can have the significant effect. Although in the above embodiment TiN, crystal grain promotion layer, of 0.1 μm is formed, the crystal grain promotion layer does not have to be a uniform layer. Namely, even when it is 1 to 5 nm thick and grain-shaped or island-shaped without being combined, the crystal grain promotion function can be obtained. It is found

that the same function or effect as TiN can be obtained in Ti, TiC, Ta, TaN and TaC. It is desirable that the dielectric film is of SiO₂ or organic material. The organic material may be formed porous.

5 The metal film of Ti, Ta or Cu may be made by using vapor deposition, laser annealing that heating is instantly conducted to melting point of metal film to be formed, plating or chemical vapor deposition, other than sputtering.

10 Although the invention has been described with respect to the specific embodiments for complete and clear disclosure, the appended claims are not to be thus limited but are to be construed as embodying all modifications and alternative constructions that may occur to one skilled in the art which
15 fairly fall within the basic teaching herein set forth.